Cuboidal Supraparticles Self-Assembled from Cubic CsPbBr₃ Perovskite Nanocrystals

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ABSTRACT: Colloidal CsPbBr₃ nanocrystals (NCs) have emerged as promising candidates for various opto-electronic applications, such as light-emitting diodes, photodetectors, and solar cells. Here, we report on the self-assembly of cubic NCs from an organic suspension into ordered cuboidal supraparticles (SPs) and their structural and optical properties. Upon increasing the NC concentration or by addition of a nonsolvent, the formation of the SPs occurs homogeneously in the suspension, as monitored by in situ X-ray scattering measurements. The three-dimensional structure of the SPs was resolved through high-angle annular dark-field scanning transmission electron microscopy and electron tomography. The NCs are atomically aligned but not connected. We characterize NC vacancies on superlattice positions both in the bulk and on the surface of the SPs. The occurrence of localized atomic-type NC vacancies—instead of delocalized ones—indicates that NC–NC attractions are important in the assembly, as we verify with Monte Carlo simulations. Even when assembled in SPs, the NCs show bright emission, with a red shift of about 30 meV compared to NCs in suspension.

Over the past decade, a lot of research has been devoted to the properties and improvement of hybrid organic–inorganic perovskite materials (e.g., CH₃NH₃PbX₃, X = Cl, Br, or I) for solar cell applications, which has led to an increase in efficiency from 3.8% to over 20%. This success has motivated researchers all over the world to study other types of perovskite materials and extend their optoelectronic applications to photodetectors, light-emitting diodes, and two-photon pumped gain media for lasers. Recently, cesium lead halide (CsPbX₃) perovskite nanocrystals (NCs) have been reported as a new and promising branch of perovskites. These colloidal NCs can be synthesized with a facile hot injection method and possess bright emission and readily tunable optoelectronic properties. In comparison to the hybrid CH₃NH₃PbX₃ compounds, the all-inorganic CsPbX₃ has a better temperature stability. Moreover these NCs show narrow emission peaks with a full width at half-maximum (fwhm) of 50–80 meV and exhibit quantum yields of up to 90% without any extra passivation of the NC surface. The emission can be tuned over the full visible spectrum by varying the composition of the X-anionic sublattice, either directly or by postsynthetic anion-exchange reactions.

NCs with low polydispersity and high photoluminescence (PL) quantum yields are ideal building blocks for larger structures with predesigned opto-electronic functionalities. In...
these superlattices, new properties can arise from the electronic and/or magnetic coupling between constituent NCs. For example, the new functionality that arises from the ordered structures of NCs has been used already in magnetic devices\(^\text{16}\) and for electronic applications.\(^\text{17}\) Here, we report on the self-assembly of cubic-shaped CsPbBr\(_3\) NCs into cuboidal supraparticles (SPs), each SP consisting of several hundreds to thousands of NCs. Using in situ X-ray scattering measurements, we show that the SPs nucleate and grow in the solution upon increasing NC concentration or by addition of a nonsolvent. The structure of the superlattices has been studied with (scanning) transmission electron microscopy [(S)TEM], electron diffraction (ED), and high-angle annular dark-field (HAADF) STEM tomography. We show that the NCs form a simple cubic lattice and are atomically aligned but not connected. We observed empty superlattice positions, that is, vacancies similar to those in an atomic crystal, in the bulk as well as on surface lattice positions. Vacancies positioned at the lattice sites should be distinguished from delocalized vacancies predicted for the hard-cube model.\(^\text{18}\) Our results thus indicate that attractive interactions between the constituent NCs are important in the self-assembly process. We studied the PL of individual SPs with confocal microspectroscopy. Despite the fact that exciton energy transfer can occur, the SPs still show a bright PL, red-shifted by 30 meV compared to a diluted NC solution.

\section*{RESULTS AND DISCUSSION}

\textbf{Formation of Cuboidal SPs.} In general, two methods can be used to induce aggregation of NCs in solution: (1) solvent evaporation, also shown by Kovalenko and Bodnarchuk,\(^\text{19}\) which increases the NC concentration, and (2) addition of antisolvent, increasing the potential energy per NC. We used methyl acetate as an antisolvent because it is one of the few polar solvents that does not damage the NCs.\(^\text{20}\) The use of other polar media, such as methanol, ethanol, acetone, and acetonitrile, as an antisolvent leads to dissolution of the individual NCs. Both methods give identical SPs, however larger SP sizes are obtained by solvent evaporation (see the Supporting Information, Figure S4). The drawback of solvent evaporation is that the SP formation takes rather long (>1 month at room temperature). When we consider self-assembly in a reasonable lab-time period of hours, care has to be taken to operate the assembly under nearly reversible conditions, as fast kinetics can lead to disordered, nonequilibrium structures. Indeed, we observed large clusters with an irregular shape and disordered structure on the NC length scale, when a relatively high amount of antisolvent was added (see the Supporting Information, Figure S5) to speed up the SP formation. The structures that we will discuss were obtained under nearly reversible conditions. A more detailed description of the parameters, which were used during the self-assembly process, can be found in the Supporting Information.

Figure 1 shows self-assembled NC SPs in a series of HAADF-STEM images with increasing magnification. A low magnification image is presented in Figure 1a. The SPs appear as large white cuboids on a dark background. At higher magnifications, it becomes clear that each SP consists of several hundreds of NCs (Figure 1b–d). Moreover, the columns of NCs are clearly distinguished at the highest magnification image (Figure 1d). The degree of ordering inside the SP is apparent through the observed contrast in the HAADF-STEM images, hinting toward a simple cubic packing of the NCs inside the cuboidal SP. The NCs are clearly not atomically...
attached as has been observed for superlattices of PbSe NCs, but they are still separated from each other most likely by their oleate and oleylamine ligands. The alignment must hence be related to the dense cubic stacking of the NC building blocks. The contrast in between the particles seems to be blurred out slightly, which we tentatively ascribe to some positional and rotational disorder of the NCs inside the SP. More images of the formed SPs can be found in the Supporting Information, Figure S3. Although cubic particles can theoretically achieve high packing fractions up to 100%, they are not often encountered in nature. In the CsPbBr₃ SPs discussed here, cubic symmetry emerges over three length scales: on the scale of the atomic lattice, the NC building blocks in the supra particle superlattice, and the cuboidal shape of the SP itself.

To properly quantify the rotational freedom of the CsPbBr₃ NCs inside the SPs, we performed ED experiments on the SP presented in Figure 1. The selected area ED pattern is shown in Figure 2a. The presence of well-defined diffraction spots, instead of powder rings, already indicates that the particles are (nearly) atomically aligned, with the three [001] type zone- axes directed in the principal directions of the NC superlattice (i.e., one of the NC {100} facets pointing upward). The diffraction pattern can be indexed up to reflections from the {420} atomic planes. In Figure 2b, azimuthal traces at the constant scattering vector q are presented for reflections from the {100}, {110}, {200}, and {220} planes (depicted in red, blue, green, and yellow, respectively). The 12 peaks are fitted with Gaussian functions to obtain an average fwhm of 9.6° ± 1.4°, which reflects an upper limit of their in-plane rotational freedom. The high degree of atomic alignment of the NCs inside the SPs originates from the densest possible stacking of NC building blocks with a uniform cubic shape and size.

It is also relevant to characterize the structure of the NC assemblies formed in the suspension before they are scooped and dried on a transmission electron microscopy grid. Therefore, we performed in situ transmission small/wide-angle X-ray scattering (SAXS, WAXS) measurements on pure stable NC suspensions and on suspensions in which the NC assembly was initiated by adding methyl acetate as an antisolvent. A schematic of the experiment is shown in Figure 3a. A solution of NCs is added into a quartz capillary with an outer diameter of 1.5 mm and placed inside a LINKAM stage, positioned at a distance of 1 m from the SAXS detector. The WAXS detector, collecting the atomic diffraction of the NCs, was positioned at the inlet of the vacuum tube holding the SAXS detector. The SAXS and WAXS patterns of a stable suspension are shown in Figure 3b. The SAXS region (light blue) only shows form factor scattering from dispersed NCs. The SAXS signal (dark blue) shows sharp diffraction peaks originating from the atomic perovskite lattice of the NCs. A sample with a volume fraction of 0.2 methyl acetate was measured after 3 days of incubation and is presented in Figure 3c. The SAXS pattern shows clear Bragg peaks at positions of 0.55, 0.78, 1.1, and 1.23 nm⁻¹. Their relative peak positions correspond to 1:√2:2:√5, which corresponds to scattering from the {100}, {110}, {200}, and {210} lattice planes of a simple cubic lattice of NCs. The measured NC–NC distance inside the SPs in suspension is determined to be 11.4 ± 0.1 nm, corresponding to NCs still separated by their oleate and oleylamine ligands. The fwhm of the {100} reflection is 0.378 nm⁻¹, which corresponds to spatial extension of the periodic order of roughly 166.3 nm. The SP formation is reversible, as the constituent NCs readily redisperse upon diluting the solution of SPs (see the Supporting Information, Figure S6).

Recent work by Bertolotti et al. also showed that the CsPbBr₃ NCs have the tendency to stack in solution, similar to concentrated solutions of perovskite nanoplatelets. They argue that the stacking direction occurs most likely in the {100} NC direction, similar to the NCs in the SPs in this work. Others have also found that these perovskite NCs readily assemble into one-dimensional chains by addition of molecular clusters or through ligand–solvent interactions. Such stacks could hence be a precursor phase in the formation of the SPs presented in this work. However, this remains to be verified by performing in situ time-resolved SAXS and WAXS
on the self-assembly process, which is beyond the scope of this article.

**NC Vacancies inside a SP Studied with HAADF-STEM Tomography and Monte Carlo Simulations.** To study the geometric structure of the NCs inside the SPs in more detail, a single SP was studied using HAADF-STEM tomography. A series of projection images was acquired with an angular range of $-70^\circ$ to $78^\circ$ and a tilt increment of $2^\circ$. Orthogonal to that, a second tilt series was acquired from $-76^\circ$ to $76^\circ$ to reduce the missing wedge of tilt angles to a missing pyramid. Using this dual tilt series as an input for several mathematical reconstruction algorithms, we found that the quality of the reconstruction was greatly improved compared to the reconstruction that was obtained from just using the one tilt series.

A visualization of the acquired tomogram is presented in Figure 4a. Unfortunately, we are not able to clearly resolve all individual NC positions inside the SP; the missing pyramid of tilt angles smears out the information in the $z$-direction of the cuboidal SP. Nonetheless, the orthoslices taken through the center of the structure in Figure 4b–d show interesting features. In all three orthogonal directions, we observe empty places on lattice positions. They appear to be localized single point vacancies and vacancy clusters. From the tomogram, we are able to estimate the volume fraction of vacancies in the bulk of the SP of $3.3\%$, assuming cubic symmetry of the NC lattice (see the Supporting Information for more details). Furthermore, we also performed secondary electron-STEM to acquire information on the surface of the SPs, which can be found in the Supporting Information (Figures S7 and S8). Also, on the surface of the SPs, we observe localized NC vacancies and groups of missing NCs. We estimate that the amount of missing surface NCs roughly equals $5.5 \pm 1.5\%$.

Interestingly, the localized nature of the NC vacancies is in contrast to previous Monte Carlo and molecular dynamics simulations for hard cubes, that is, cubes that interact only through excluded volume interactions. Smallenburg et al. showed that for hard cubes, vacancies manifest themselves as a finite-length chain of particles along one of the principal axes in the crystal. The vacancy is thus spread over many lattice positions. More specifically, in this hard-cube system, the equilibrium concentration of vacancies is very high (up to $6\%$) because of the fact that creating a vacancy provides additional free volume for multiple nearby particles, which increases the entropy of the crystal. Note that in a system of attractive cubes, Rossi et al. observed localized NC vacancies, where colloidal cubes of roughly 1 micron are self-assembled into simple cubic lattices through addition of a depletant, which induces effective attractions between the cube facets. As such, we propose that the localization and grouping of NC vacancies in our experiments point toward attractive interactions between the NCs during the self-assembly. Specifically, for sufficiently large NC–NC attractions, the free energy will be minimized when the particles in the vicinity of the vacancy are sitting on regular lattice positions, as the cost of breaking cohesive bonds will dominate over the entropic gain associated with vacancy delocalization.

To understand the observed localization of the vacancies, we performed Monte Carlo simulations of cubes in the NVT ensemble, where $N$ (the number of particles), $V$ (the volume), and $T$ (the temperature) are fixed, and the cube–cube attraction can be changed from zero to several times the thermal energy. As a minimal model that captures the repulsive core and attraction, we model the interactions between the particles in the vicinity of the vacancy as a system of attractive cubes.

$$
\text{Energy} = k_b T \left( T - T_N \right) \left( N - N_N \right)
$$

where $k_b$ is the Boltzmann constant, $T$ is the temperature, $T_N$ is the number of particles, $N$ is the volume, and $N_N$ is the equilibrium concentration of vacancies. The vacancy is thus spread over many lattice positions. More specifically, in this hard-cube system, the equilibrium concentration of vacancies is very high (up to $6\%$) because of the fact that creating a vacancy provides additional free volume for multiple nearby particles, which increases the entropy of the crystal. Note that in a system of attractive cubes, Rossi et al. observed localized NC vacancies, where colloidal cubes of roughly 1 micron are self-assembled into simple cubic lattices through addition of a depletant, which induces effective attractions between the cube facets. As such, we propose that the localization and grouping of NC vacancies in our experiments point toward attractive interactions between the NCs during the self-assembly. Specifically, for sufficiently large NC–NC attractions, the free energy will be minimized when the particles in the vicinity of the vacancy are sitting on regular lattice positions, as the cost of breaking cohesive bonds will dominate over the entropic gain associated with vacancy delocalization.

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upon introducing a slight attraction between the cubes, energy transfer. We found that CsPbBr₃ SPs deposited on a the NCs in the solid enhances the probability of nonradiative reduced PL quantum yield, as exciton energy transfer between particles. For slightly larger attractions, attraction strength between the particles, we observe the defect over a couple of lattice sites. Thus, already for a moderate

Often, quantum dot (QD) solids su

in Figure 4.

(Figure 5c), similar to the tomography experiments presented in Figure 6.

Optical Properties of SPs Compared to NCs in Solution. Often, quantum dot (QD) solids suffer from a reduced PL quantum yield, as exciton energy transfer between the NCs in the solid enhances the probability of nonradiative energy transfer. We found that CsPbBr₃ SPs deposited on a glass slide remained strongly emissive, with a PL quantum yield of 26%, as determined with an integrating sphere. This is good news for several opto-electronic applications requiring a QD solid as the optically active material. To study the optical properties of individual SPs in more detail, we performed confocal microspectroscopy. A dispersion containing the SPs in concentrated solutions of CsPbBr₃ NCs and solid-state films.  This effect is also shown in concentrated solutions of the cubic CsPbBr₃ NCs and is reversible upon diluting the NC dispersion. In our time-resolved PL experiments (Figure S14), we observe that the shift in the PL peak position occurs faster than the time resolution of our setup (∼250 ps), which is probably faster than the exciton energy transfer. This indicates that energy transfer is not the cause for the observed red shift. We must conclude here that although the red shift occurring in CsPbBr₃ NC SPs is well-established, its main cause remains an outstanding question.

In summary, we have shown that CsPbBr₃ nanocubes self-assemble into SPs that have an overall cuboidal shape. In the superlattices, the NCs are atomically aligned but separated by the capping ligands. Vacancies form real point defects instead of being delocalized, indicating that attractive interactions play a role in the self-assembly process. The SPs show a bright PL, red-shifted by 30 meV with respect to that of individual NCs. The fact that the obtained SPs are still highly emissive makes them promising candidates for opto-electronic applications. For example, they possibly can be used as microcavities to study the confinement of the PL and lasing inside SPs.
ASSOCIATED CONTENT

Supporting Information
The Supporting Information is available free of charge on the ACS Publications website at DOI: 10.1021/acs.jpcc.8b02699.

Synthesis of CsPbBr3 NCs, purification of the NCs, self-assembly of the NC solutions into SPs, transmission SAXS and WAXS experiments, and optical spectroscopy (PDF)

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Notes
The authors declare no competing financial interest.

ACKNOWLEDGMENTS

The authors thank Dr. Rajeev Dattani and Jacques Gorini from the ID02 beamline of the ESRF for their excellent assistance during the X-ray scattering experiments. We also thank Carlo van Overbeeke, P. Tim Prins, and Federico Montanarella for their support during the synchrotron experiments. The authors gratefully acknowledge Prof. Dr. Alfons van Blaaderen for fruitful discussions. D.V. acknowledges funding from NWO-CW TOPPUNT “Superior superstructures.” J.J.G. acknowledges the joint Debye and ESRF graduate programs for the financial support. H.V. gratefully acknowledges the financial support by the Flemish Fund for Scientific Research (FWO grant 1S32617N). S.B. acknowledges the financial support from the European Research Council (ERC Starting grant #335078-COLOURATOMS). Y.Z. acknowledges the financial support from the European Union’s Horizon 2020 research and innovation program, under the Marie Skłodowska-Curie grant agreement #665501 through a FWO [PEGASUS].

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